

# Dry Sliding Wear Mechanism of Spark Plasma Sintered $\text{Si}_3\text{N}_4/\text{SiC}$ Composites on Steel

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## Article Information

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In tribological applications, the use of engineering ceramics has been increasing. Silicon nitride ceramics are considered for use in many mechanical applications, especially for rolling contact systems (e.g. ball bearings) or for reciprocating engine parts (e.g. valves, valve seats, valve guides, and various industrial wear parts) [1-9]. In order to improve mechanical properties of monolithic  $\text{Si}_3\text{N}_4$  ceramics, a second phase like platelets, whiskers or particles are added to the  $\text{Si}_3\text{N}_4$  matrix [2, 10, 11]. Silicon nitride reinforced with silicon carbide is a potentially important component in a number of ceramic composites [12]. Reinforcement of ceramics by SiC is often applied to abrasive, erosive or impact wear [3]. The densification of  $\text{Si}_3\text{N}_4$ , commonly entails the use of  $\alpha\text{-Si}_3\text{N}_4$  powders plus rare earth oxide additives, and temperatures as high as 1700 to 1800 °C for promoting the liquid-phase sintering. Rare earth (RE) oxide sintering additives are considered as ideal additives for  $\text{Si}_3\text{N}_4$  due to their high melting point and due to the fact that they control the  $\alpha \rightarrow \beta$  phase transformation rate, the grain growth anisotropy and the aspect ratio of the  $\beta\text{-Si}_3\text{N}_4$  grains [2, 9]. The SPS technique can enhance the sinterability of  $\text{Si}_3\text{N}_4$  producing dense materials with

In the present work,  $\text{Si}_3\text{N}_4$ -based composites were characterized by a relatively low friction coefficient in unlubricated sliding against a 100Cr6 steel ball.  $\text{Si}_3\text{N}_4$ -based composites containing different amounts of SiC were produced by spark plasma sintering at 1650 °C for 5 min with a uniaxial pressure of 40 MPa in a nitrogen atmosphere. In the wear test, the load, total distance and rotating speed were selected as 60 N, 120 m and 500 rpm, respectively. The wear tracks were examined by scanning electron microscopy (SEM) to estimate the wear mechanism. The results were evaluated using the friction coefficient-distance diagram.  $\text{Si}_3\text{N}_4$ -based composites showed significant decrement in wear by improving toughness with SiC powder reinforcement.

controlled grain growth and phase transformation at reduced temperatures [2, 9, 13, 14]. Tribological materials are also required to possess higher mechanical properties like structural materials. The mechanical properties, such as strength and fracture toughness are closely related to their microstructures [4]. Tribological behavior of ceramic materials is strongly dependent on contact load, sliding speed, temperature, humidity, contact geometric configuration, lubricant, as well as microstructure of materials [2, 6, 15]. The purpose of this study is to investigate the influence of SiC particles added to  $\text{Si}_3\text{N}_4$  matrix on the wear behaviour of  $\text{Si}_3\text{N}_4/\text{SiC}$  composite.

## Experimental Procedure

$\alpha\text{-Si}_3\text{N}_4$  powder (UBE Industries, SN-E10, 38 wt.-% N, > 2 wt.-% O, > 0,2 wt.-% C), SiC powder (H.C. Starck, grade UF-25, > 28,5-29,5 wt.-% C, > 2,5 wt.-% O) were used as powder precursors. High purity AlN (H.C. Starck, grade C) and  $\text{Y}_2\text{O}_3$  (H.C. Starck, grade C) were used as sintering additives. The composition of the starting powder mixtures was given in Table 1. Powder batches were produced by wet ball milling with ethanol in a polyurethane bottle for

16 h. The slurries were dried at 100 °C in 5 h, and then sieved to a particle size < 185  $\mu\text{m}$ . The powder precursors were loaded in a cylindrical graphite die with an inner diameter of 50 mm and uniaxially pressed into green compacts by using a hydraulic pressure of 40 MPa. The green compacts were sintered in DR SINTER SPS (spark plasma sintering) -7.40 MK by applying 40 MPa pressure for 5 minutes at 1650 °C in nitrogen atmosphere. The heating rate was 100 °C  $\text{min}^{-1}$ . The temperature was measured by means of an optical infrared thermometer focused on the graphite die surface.

Relative density of the samples after sintering was determined by Archimedes method. The crystal phases of the samples were investigated by X-ray diffractometry (XRD) using X'Pert Pro MRD Panalytical diffractometer. Spark plasma sintered samples were cut and polished to a 1  $\mu\text{m}$  dia-

Sample	$\text{Si}_3\text{N}_4$	SiC	AlN	$\text{Y}_2\text{O}_3$
SSC10	80	10	5	5
SSC30	60	30	5	5

Table 1. Composition of starting powder mixtures [wt.-%]

mond finish. The microstructures of the samples were characterized by scanning electron microscopy (SEM) (JEOL JSM-7000 F). Hardness of the samples was tested by a micro Vicker tester (Struers Duramin A-300) with an applied load of 2 kg. The strength was measured by three-point bending. For tribological characterization, wear tests were performed under dry sliding condition and wear characteristics were then determined based on the worn surfaces. The wear behavior of the composites was studied by unlubricated “ball-on-disc” type tribometer (Nanovea) against a polished 100Cr6 steel ball at room temperature. Related friction and wear test parameters were listed in Table 2. The wear tracks were examined by SEM to estimate the wear mechanisms.

**Results and Discussion**

XRD patterns of 1650 °C under 40 MPa pressure for 5 min spark plasma sintered samples SSC10 and SSC30 were shown in Figure 1. The main phases were α-Si<sub>3</sub>N<sub>4</sub>, β-Si<sub>3</sub>N<sub>4</sub>, SiC and SiAlON in both composite materials. Also, Y<sub>2</sub>O<sub>3</sub> could be detected in SSC10 composite.

Mechanical properties of SSC10 and SSC30 composites sintered at 1650 °C under 40 MPa for 5 minutes were given in Table 3. Fracture toughness value of SSC30 was 6.0 MPa × m<sup>1/2</sup>, while the fracture toughness value of SSC10 was 5.8 MPa × m<sup>1/2</sup>. Higher toughness exhibited better wear resistance. The specimen was more resistant to wear when it had a high bending strength and hardness. The fracture toughness of Si<sub>3</sub>N<sub>4</sub>-based composites played an important role for wear behavior [15, 16]. The fracture toughness of silicon nitride was influenced by the microstructure characteristics (grain diameter, aspect ratio of grains, amount of elongated grains). The composite showed decrement in wear by improving toughness with SiC powder reinforcement. Tarko et al. reported that the wear rate of

Si<sub>3</sub>N<sub>4</sub> ceramic composite decreased with the increasing fracture toughness. The hardness values of SSC10 and SSC30 were 21.4 GPa and 22.5 GPa, respectively. This can be explained by the presence of harder SiC particles in the Si<sub>3</sub>N<sub>4</sub> matrix [2].

Because of the high density of the samples, the pores were not observed from the micrographs given in Figure 2. SiC particles in the powder composite hindered the α-β-Si<sub>3</sub>N<sub>4</sub> phase transformation and the growth of β-Si<sub>3</sub>N<sub>4</sub> grains into the Si<sub>3</sub>N<sub>4</sub> matrix.

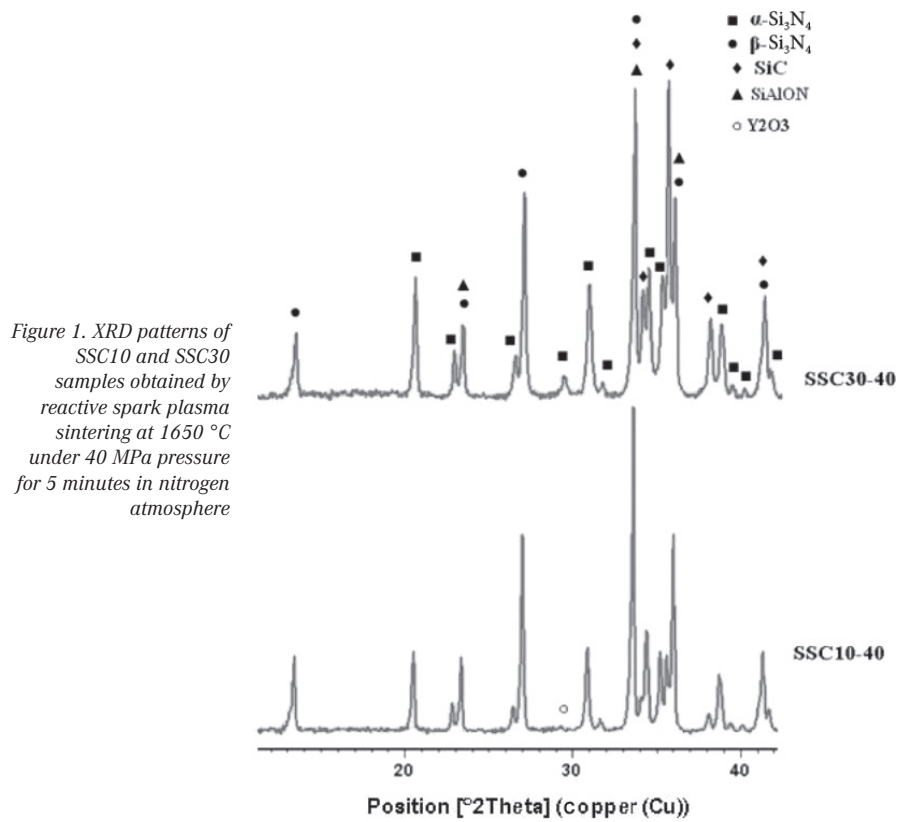


Figure 1. XRD patterns of SSC10 and SSC30 samples obtained by reactive spark plasma sintering at 1650 °C under 40 MPa pressure for 5 minutes in nitrogen atmosphere

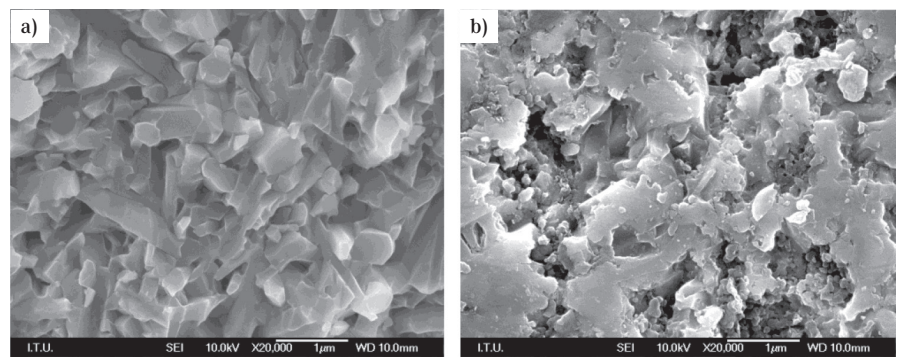


Figure 2. Microstructures of the composites spark plasma sintered at 1650 °C under 40 MPa for 5 minutes, a) SSC10, b) SSC30

Counterpart material	100Cr6
Counterpart hardness	65 HRC
Friction velocity	0.1 m × s <sup>-1</sup>
Normal load	60 N
Total sliding distance	120 m
Surrounding atmosphere	ambient air
Lubrication method	none

Table 2. Parameters used for the wear tests

Sample	Hardness	Elastic modul, E [GPa]	Theoretical density [%]	Density [g × cm <sup>-3</sup> ]	Bending strength [MPa]	Fracture toughness [MPa × m <sup>1/2</sup> ]
SSC10	21.4	335	99.6	3.28	317	5.8
SSC30	22.5	379	99.3	3.27	367	6.0

Table 3. Mechanical properties of SSC10 and SSC30 composites sintered at 1650 °C under 40 MPa for 5 minutes

The friction coefficients of  $\text{Si}_3\text{N}_4/\text{SiC}$  composites against 100Cr6 steel ball as a function of distance were shown in Figure 3. The SSC30 ceramic composite showed a lower friction coefficient than SSC10 ceramic composite. The friction coefficient values of SSC10 were nearly the twice of the friction coefficient values of SSC30. The addition of SiC particles to  $\text{Si}_3\text{N}_4$ -based composites showed a lower friction coefficient during sliding.

The wear occurs by microfracture under dry conditions. The tribochemical

wear mechanism was often accompanied by material transfer and adhesion, mostly from softer steel to the ceramic specimen. The presence of tribochemical films on ceramic and metals under sliding or fretting conditions were reported many times [3, 4, 6, 8, 15]. Although their strong dependence on environmental and operating conditions was confirmed, details of the mechanisms and chemical reactions are still not understood. The situation is complicated by the many influential parameters and variations in test conditions

[17]. The wear of  $\text{Si}_3\text{N}_4$  as a result of the adhesion with the steel ball followed by the pull-out of the  $\text{Si}_3\text{N}_4$  grains. The microstructure of silicon nitride ceramics has a significant effect on their mechanical and tribological properties. The grain morphology, in particular, is a significant factor for controlling wear resistance [5]. Under dry frictional conditions, the wear of  $\text{Si}_3\text{N}_4$  ceramic in the ceramic steel sliding contacts was thought to be mainly caused by the adhesion between the rubbing surfaces and the microfracture of the ceramic [18]. The wear microstructures of the samples SSC10 and SSC30 were shown in Figure 4. The wear tracks in SSC10 specimen were deep, intense and wider compared to SSC30 specimen. As the SiC powder content increased to 30 wt.%, material removal via delamination was seen to be less.

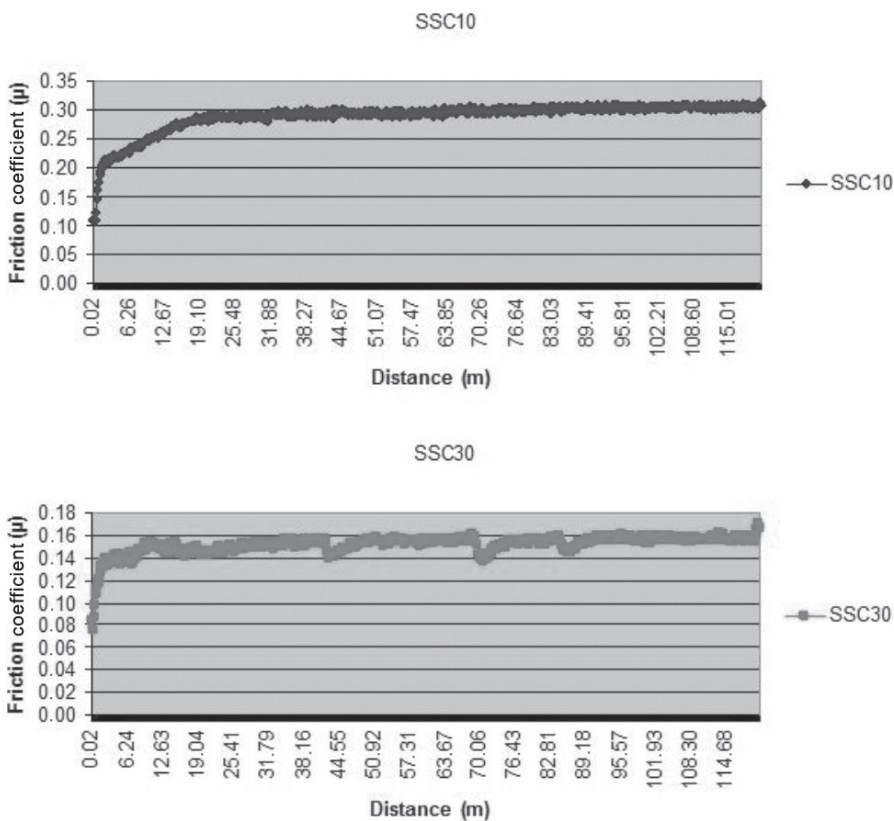


Figure 3. Friction coefficients of  $\text{Si}_3\text{N}_4/\text{SiC}$  composites against 100Cr6 steel ball as a function of distance

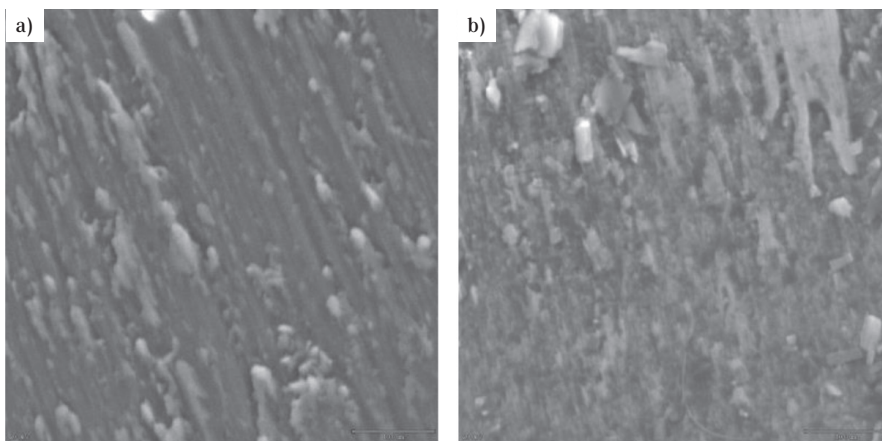


Figure 4. Wear microstructures of the spark plasma sintered samples at 1650 °C under 40 MPa for 5 minutes, a) SSC10, b) SSC30

## Conclusions

The following conclusions could be listed from the results obtained in this study:

- Because of the reinforcement effects of SiC, the obtained composites showed increased mechanical properties such as high hardness and fracture toughness. The samples had relative densities higher than 98%.  $\alpha\text{-Si}_3\text{N}_4$ ,  $\beta\text{-Si}_3\text{N}_4$ , SiC and SiAlON were the main phases. SiC particles in the powder composite hindered the  $\alpha\text{-}\beta\text{-Si}_3\text{N}_4$  phase transformation and the growth of  $\beta\text{-Si}_3\text{N}_4$  grains into the  $\text{Si}_3\text{N}_4$  matrix.
- $\text{Si}_3\text{N}_4$ -based composites were characterized by a relatively low friction coefficient in unlubricated sliding against 100Cr6 steel ball.
- The adding of SiC particles increased the fracture toughness of  $\text{Si}_3\text{N}_4$ -based composites. The fracture toughness of  $\text{Si}_3\text{N}_4$  played an important role in their wear behavior. The addition of SiC particles to  $\text{Si}_3\text{N}_4$ -based composite increased its wear resistance.
- The wear tracks in SSC10 specimen were deep, intense and wider compared to SSC30 specimen. As the SiC powder content increased to 30 wt.%, material removal via delamination was seen to be less.

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## Abstract

**Trockengleit-Verschleißmechanismus von funkenplasmagesinterten Si<sub>3</sub>N<sub>4</sub>-SiC-Kompositen gegenüber Stahl.** In dem diesem Beitrag zugrundeliegenden Forschungsarbeiten wurden Si<sub>3</sub>N<sub>4</sub>-basierte Kompositwerkstoffe bei einem relativ niedrigem Reibungskoeffizient unter geschmiertem Gleiten gegen einen Ball aus 100Cr6 Stahl charakterisiert. Die Si<sub>3</sub>N<sub>4</sub>-basierten Komposite mit verschiedenen Mengen von SiC wurden durch Plasmafunkenintern bei einer Temperatur von 1650 °C für 5 min bei einem einachsigen Druck von 40 MPa in einer Stickstoffatmosphäre hergestellt. In dem Verschleißtest wurden die Last, der Gesamtabstand und die Rotationsgeschwindigkeit mit 60 N, 120 m und 500 U × min<sup>-1</sup> gewählt. Die Verschleißspuren wurden mittels Elektronenmikroskopie untersucht, um den Verschleißmechanismus abzuschätzen. Die Ergebnisse wurden mittels eines Reibungskoeffizient-Entfernungs-Diagramms ausgewertet. Die Si<sub>3</sub>N<sub>4</sub>-basierten Komposite zeigten eine signifikante Abnahme des Verschleißes mit verbesserter Zähigkeit bei einer Verstärkung mit SiC-Pulver.

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